

1. Introduction

A key component of nanomedicine and nanobiotechnology is the use of nanomaterials. Antibiotic-resistant microorganisms may be able to benefit from this new strategy [1]. Because of their unique physical, chemical, mechanical, and magnetic properties, nanoparticles (NPs) have been shown in multiple studies to be therapeutic agents for treating infections caused by bacteria. As a result, new bactericidal agents are being developed using NPs in clinical settings [2-3]. An efficient prevention of bacterial growth can be achieved by using nanoscale materials that can penetrate into bacterial cells and produce deadly oxygen ion radicals that damage the cell membranes of germs [4-7]. Using spinelstructured ferrite nanoparticles in magnetic resonance imaging, magnetic storage devices, biotechnology, electronic devices, magnetic medication delivery, and other fields has made a significant influence on science researchers [8-10]. For many applications, magnetic nanoparticles require super-paramagnetic properties at room temperature because of their huge surface-to-volume ratio [11-14]. MnFe2O⁴ and nanoparticles have the highest surface area, the highest saturation magnetization, the highest mechanical hardness, and the best chemical stability of all spinel ferrites [15]. It has been found in the literature that manganese ferrite nanoparticles may be manufactured using a variety of processes, including classic ceramic methods, oxalate method, sol–gel method, chemical coprecipitation and mechanical ball milling [16]. There are numerous uses for the sol–gel process, which is one of the most widely used procedures. Scientists used this method to create some of the world's lightest materials and some of the toughest ceramics. The spin or dip coating of thin films on a substrate using the sol–gel technique has a wide range of applications. It's possible to make glass objects with unique features that can't be achieved any other way. The sol–gel method has used a variety of surfactants up to this point, but there is no information on the effects of WF and PF as surfactants. For the production of manganese ferrite, WF and PF were employed as surfactants. A and B samples of $MnFe₂O₄$ and Mn-Zn-Fe2O⁴ /TiO² were studied for their structural, morphological properties (XRD analysis, FT-IR spectroscopy, Filed emission scanning electron microscopy (FE-SEM)). Particle size and shape, morphology, and chemical composition all play important roles in determining the samples' physical qualities, however these factors cannot always be controlled during the nanoparticle production process [16]. For Manganese-Zinc-Ferrite (Mn-Zn-Fe2O4) nanocomposite prepared through ceramic methods like sol-gel, the composition, cationic distribution and thermal behavior of oxygen are very often in-ter related, leading to an important dependence of the mi-cross

structure and therefore its physical properties, with the atmosphere and heat treatment temperatures [17].

 Despite this, sol-gel synthesis is a viable option with significant economic and technological advantages as well as excellent reproducibility, and it can produce monodisperse nanoparticles with good crystallinity when the synthesis process is conducted with the right control [18-19]. By using XRD analysis (XRD-6000/Japan), the (Mn-Zn-Fe2O4) nanocomposite was characterized and the findings were compared to the JCPDS card data (Joint Committee on Powder Diffraction Standards). In Iraq's Nanotechnology and Advanced Materials/Materials Research Department/Ministry of Science and Technology, XRD measurements have examined the orientation of $(Mn-Zn-Fe₂O₄)$ nanocomposite grown samples. In Iran-Mashhad, "(Tescan Mira3 FE-SEM-Czechia)" employed FE-SEM to analyze the morphology and particle size of (Mn-Zn-Fe2O4). Fourier transformed infrared spectrum (FTIR) was utilized to study the functions group and vibrations modes utilizing a potassium bromide (KBr) beam splitter and a mid IR triglycine sulfate (TGC) detector for the polarized IR reflectance. The magnetic characteristics were examined utilizing an MPMS XL, Quantum Design SQUID magnetometer by VSM.

2. Experimental work

2.1 The materials used

The materials used in the preparation of the ferrite compound by sol-gel auto combustion method are listed in table 1 with their full description.

| Compounds | Chemical formula | Mol. mass $(g mol-1)$ | Purity $\%$ | Origin |
|--------------------------|---------------------|--------------------------|-----------------------|------------|
| Titanium oxide | TiO2 | 79.87 | 99.00 | USA |
| (III) Iron nitrate | $Fe(NO3)3.9H2O$ | 404.00 | 97.00 | India |
| Citric acid | $C_6H_8O_7.H_2O$ | 210.14 | 99.00 | USA |

Table (1): The chemical materials and their properties

2.2 Preparation method of the Mn-Znferrites and Mn-Zn-ferrites /TiO² nanocomposite

All metal nitrate and citric acid were dissolved in 15 ml of distilled water. All of them are combined in a glass beaker to form a complete solution, which is thoroughly mixed at room temperature with a magnetic stirrer at high speeds for a short time until the solution becomes smooth and slimy red-colored, as shown in figure (1a). Ammonia solution was steadily injected into the mixed solution in the form of drips with continual stirring to regulate its pH until it reached the value of (7), resulting in a dark brown tint as seen in the picture (1 b). To achieve homogeneity, stir the components together for half an hour at room temperature. Gradually raise the temperature until it reaches 90 °C, stirring constantly until the gel form is reached. The size of the solution in the beaker glass be less with high viscosity, and after (30 minutes), the solution viscosity is very high, resulting in the beginning of gel formation on the surface of the solution, particularly in the middle, and then all of the solution turns to gel, and even at this point, the solution is still on

the magnetic stirrer and temperature (90 C), as shown in Figure $(1 \ c)$. The temperature decreases to room temperature after the solution turns to gel, the gel dries, and the hue changes to dark brown. Where the weight of the gel is placed in a glass beaker using a sensitive balance, and then placed in the oven at 150°C for three hours to dry the gel and reduce its weight as shown in Figure (1d). After evaporating some of the material and raising the temperature of the dried gel to (250°C), the dry gel began to change shape and become convex in the middle of the glass beaker after 15 minutes, as illustrated in figure (1 e). After combustion, the dried gel turns into a fine powder with a dark gray hue, signaling the start of the creation of high purity ferrite, as illustrated in the picture (1f-g). The spinel was then calcined for three hours at temperatures of (650,750, and 850 C) to improve crystallization and homogenous action distribution, and then ground to produce ZnxMn1-xFe2O⁴ ferrite Nano powders. Using the traditional ceramic method, titanium dioxide is mixed with zinc-manganese ferrite in different quantities at 500°C.

Figure (1): Photograph of (a) nitrates-citrate solution, (b) The solution after the Adding of ammonia, (c) dry gel, (d) Dry bulk temperature of 150̊C, (e) Dry bulk temperature of 250̊C and (f-g) auto combustion and become Nano powder ferrite.

2.3 Calcinations and Pellet Formation

The resulting powder is placed in a ceramic pot and then roasted at 650, 750, 850 °C for two hours to remove reaction residues such as water or carbon dioxide molecules from combustion, resulting in the necessary ferrite powder. Then 0.5 g of burnt samples and samples that were calcined at 650, 750 and 850 were taken and five drops of PVA were added as a binder to be pressed into round pellets 1 cm in diameter and 3 mm thick. This is achieved by pressing 500Pa for five minutes

to cool slowly naturally to check their dielectric

properties, as shown in figure (2).

with a hydraulic press using the dry-press method, then sintering at 400, 600 and 800 for 2 hours to thicken the samples and allow them

Figure (2): Transformation of powder to compact sample.

3. Result and discussion

3.1 XRD of the Mn-Zn-ferrites and Mn-Znferrites /TiO² nanocomposite

From figure 3 observed that at the concentration of zinc metal (x=0.45), a pure single-phase cubic spinel $Zn_xMn_{1-x}Fe_2O_4$ structure was obtained with space group (Fd-3m no. 277), crystal dimensions $(a = b = c = a$ 8.438 A°) and crystal angles ($\alpha = \beta = \gamma = 90^{\circ}$), agreed with the standard data (JCPDS 98-017- 0912) as shown in figure 3. The XRD patterns at high calcination temperatures 750 and 850 ^oC demonstrated that the distinct structural stability of obtained $Zn_xMn_{1-x}Fe₂O₄$ during the calcination process leads to crystalline growth in the same cubic $Zn_xMn_{1-x}Fe₂O₄$ structure with a high purity phase, no other impurities peaks were detected, this is an indication that all the reactant metal ions entered into the cubic ZnxMn1-xFe2O4 structure at the concentration of zinc metal $(x=0.45)$ [20]. The lattice constant value an of pure cubic Zn_xMn_{1-xFe2}O₄ ferrite nanocomposite is reduced when Zn metal added which can be explained on the basis of cations distribution, as a result of replacing

larger ionic radii of Mn^{2} cations (0.082 nm) by smaller ionic radii of Zn^{2} cations (0.074 nm) [21-22]. In order to investigate the synthesized $Zn_{0.45}Mn_{0.55}Fe₂O₄$ ferrite nanocomposite mixed with different content of TiO₂ nanoparticles and calcined at 500 $\,^{\circ}$ C, Figure 4 presents the XRD patterns of pure TiO² nanoparticles and Zn0.45Mn0.55Fe2O⁴ mixed with different ratios (20, 30, 40, 50 and 60%) of TiO₂ nanoparticles. All the obtained peaks of $Zn_{0.45}Mn_{0.55}Fe₂O₄/TiO₂$ (40:60%) nanocomposite are assigned to the characteristic peaks of two main phases which are cubic spinel Zn0.45Mn0.55Fe2O4 phase (marked with red stars) (98-017-0912) and tetragonal TiO₂ phase (Anatase) (marked with blue stars) with space group (I41/amd no.141) (98-015-4609), in addition to little cubic MnFe2O⁴ phase (marked with black stars) (98- 003-0237). The peak intensity proportion of tetragonal TiO₂ (011) to $Zn_{0.45}Mn_{0.55}Fe₂O₄$ (113) obviously increased with increase the mixed TiO² content in the structure of Zn0.45Mn0.55Fe2O4/TiO² nanocomposites

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Figure 3 XRD patterns of Zn_{0.45}Mn_{0.55}Fe₂O₄ ferrite nanocomposites at different calcination temperatures (650, 750, 850 °C).

The crystalline size of synthesized Zn0.45Mn0.55Fe2O⁴ at different calcination temperature was calculated using the Debye– Scherrer's equation, to be (27.99, 29.07 and 30.77 nm) at the temperature (650, 750 and 850 \circ C) respectively, as shown in table 2, 3, which presents the XRD calculations of synthesized ZnxMn1-xFe2O4. Observed that the crystalline size slightly increased with raising the calcination temperature and the zinc content which is the common behavior of all

Figure 4 XRD patterns of (a) pure $TiO₂$, (b), (c), (d), (e) and (f) are $Zn_{0.45}Mn_{0.55}Fe₂O₄ mixed with$ different ratios of TiO₂ (20, 30, 40, 50 and 60%) respectively.

synthesized material, attributed to the crystal growth with increase the calcination temperature and the zinc content, and this is corresponded with the measured XRD patterns where the intensity of the most intense peak (113) of pure $Zn_{0.45}Mn_{0.55}Fe₂O₄$ is significantly increases and the pattern appears narrower with the increasing of Zn content [2**Error! Bookmark not defined.**]. On other side, the crystalline size of $TiO₂$ nanoparticles is $(10.76$ nm).

Table 2 XRD calculations of $Zn_xMn_{1-x}Fe₂O₄$ with different content of zinc metal (x=0, 0.15,

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3.2 FE-SEM of the Mn-Zn-ferrites and Mn-Znferrites /TiO2 nanocomposite

The particle size got larger when the calcination temperature was raised, as shown in the FE-SEM image. FE-SEM images of manganese ferrite nanoparticles after calcination at (650, 750, and 850) \degree C are shown in Figures 5 (A-C) At 650 oC, spherical ferrite particles agglomerated and separated significantly. At greater calcination temperatures, larger spherical and elongated particles were observed.

As shown in Figure 6 (T1-T5), FE-SEM images were utilized to examine the surface morphology and particle size of Zn0.45Mn0.55Fe2O4/TiO² nanocomposites calcined at 500 °C (T1, T2, T3, T4 and T5) All of the specimens contain a compact order of homogenous nanoparticles with a spherical form and polyhedral particles. The estimated diameters of Zn_{0.45}Mn_{0.55}Fe₂O₄/TiO₂ nanocomposites for calcined samples at 500 with relatively well-crystallized grains and average particle size less than 27.77, 43.60, 61.51, 47.29 and 72.04 nm respectively calculated by Image J Software. All data agree relatively well with the XRD results. We also note the coalescence of adjacent grains with each other due to the increase in the concentration of TiO2.

The particle size measured using FE-SEM micrographs is larger than the XRD data estimates. The XRD method is responsible for the molecular structural disruption and lattice strain generated by differing ionic radii and/or nanoparticle collection. As a result, it has a more stringent requirement, which leads to lower sizes [24].

Figure (5): FE-SEM micrographs of Zn $_{0.45}$ Mn $_{0.55}$ Fe₂O₄ Nano ferrites for (A) calcined specimen at 650 ℃ (B) at 750 ℃ and (C) at 850 ˚C.

Figure (6): FE-SEM micrographs of (a) pure TiO₂, (b), (c), (d), and (e) are $Zn_{0.45}Mn_{0.55}Fe₂O₄ mixed with$ different ratios of TiO₂ (20, 30, 40, and 50 %) respectively.

3.3 FT-IR spectrum of the Mn-Zn-ferrites and Mn-Zn-ferrites /TiO² nanocomposite

Figure 7 shows the FT-IR spectra of Zn0.25Mn0.75Fe2O⁴ ferrite nanocomposites. The asymmetric and symmetric C=O stretching modes are responsible for the bands at 1,516 and 1,600 cm-1, respectively, because metal ions coordinate the citric acid's carboxylate groups. The absorption peak at roughly 444, 567, and 665 cm-1 was clearly proven to be caused by the stretching of Fe–O, and Mn–O, ZnO bonds [25 - 26].

Analysis of the adsorbent's functional groups was carried out using the Fouriertransform infrared spectroscopy developed by BRUKER and the ALPHA model at a range of 400 to 3800 cm-1. Because each band belongs to a distinct group, the functional groups and the numbers assigned to them can be determined, as illustrated in Figure 8 [27]. Figure 6 shows the FT-IR spectrum of pure $TiO₂$ nanoparticles and $Zn_{0.25}Mn_{0.75}Fe₂O₄$ mixed with different ratios (20, 30, 40, 50, and 60 %) of TiO² nanoparticles, which was calcined at 500 °C to study the produced Zn0.25Mn0.75Fe2O4 ferrite nanocomposite. The peak at 1641 cm-1, 2344 cm -1 is indicative of the bands in the TiO₂ substance, whereas the peak at 500 cm-1 to 600 cm-1 is indicative of metal Fe-O, Mn-O and Zn-O bonds [28].

Figure (7): FT-IR spectrum of $Zn_{0.45}Mn_{0.55}Fe₂O₄$ ferrite nanocomposites at different calcination temperatures (650, 750, and 850 °C) by sol-gel method.

Wavenumber (cm-1)

Figure (8): FTIR spectrum of (A) pure TiO₂, (B), (C), (D), (E) and (F) are $Zn_{0.45}Mn_{0.55}Fe₂O₄$ mixed with different ratios of TiO₂ (20, 30, 40, 50 and 60%) respectively.

4. Conclusion

 $Mn-Zn-Fe₂O₄$ and $Zn_{0.45}Mn_{0.55}Fe₂O₄$ / TiO² NPs were successfully synthesized by a simple sol–gel method at 650, 750, and 850 $\,^{\circ}$ C. XRD results confirm pure cubic spinel crystalline structure for the sample. The average crystallite size was calculated by Debye-Scherrer formula. An examination into the possibility of producing the Mn-Zn-ferrites

(NPs) by sol-gel and Mn-Zn-ferrites $/TiO₂$ (NC) as prospective structure-tunable nanomaterials is described in this report. X-ray diffraction (XRD), field emission-scanning electron microscopy (FE-SEM), and Fouriertransform infrared (FTIR) spectroscopy were used to characterize the structural properties of the Mn-Zn-ferrites by sol-gel and Mn-Znferrites /TiO₂ (NC). They were 27 to 30 nm in

size with a cubic structure, while the Mn-Znferrites $/TiO₂$ (NC) were 10 to 30 nm in size with a cubic structure. FE-SEM images showed the synthesized the Mn-Zn-ferrites and the Mn- Zn -ferrites /TiO₂ (NC) had a nanoparticle-like structure with a particle size of 27.77, to 72.04 nm. The strong FT-IR absorption peaks at 500 and 600 cm-1 represented an Fe-O, Mn-O, and Zn-O vibration band and 1667 cm-1 represented an TiO² for the Mn-Zn-ferrites and the Mn-Zn-ferrites $/TiO₂$ (NC), respectively.

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